Thick Transparent Nanoparticle-Based Mesoporous Silica Monolithic Slabs for Thermally Insulating Window Materials

- Supporting Information -

Michal Marszewski,¹ Sophia C. King,² Yan Yan,² Tiphaine Galy,¹ Man Li,¹ Ali Dashti,¹ Danielle M. Butts,³ Joon Sang Kang,¹ Patricia E. McNeil,³ Esther Lan,³ Bruce Dunn,^{3,4} Yongjie Hu,¹ Sarah H. Tolbert,^{2,3,4} Laurent Pilon^{1,4,5*}

¹Mechanical and Aerospace Engineering Department, University of California, Los Angeles, Los Angeles, California 90095, USA

²Department of Chemistry and Biochemistry, University of California, Los Angeles, Los Angeles, California 90095, USA

³Department of Materials Science and Engineering, University of California, Los Angeles, Los Angeles, California 90095, USA

⁴California NanoSystems Institute, University of California, Los Angeles, Los Angeles, California 90095, USA

⁵Institute of the Environment and Sustainability, University of California, Los Angeles, Los Angeles, California 90095, USA

*E-mail: pilon@seas.ucla.edu

Nomenclature

а	amount adsorbed (cm ³ STP g^{-1})
B_{nh}	reference signal for normal-hemispherical transmittance measurement (%)
C_p	specific heat capacity (J $g^{-1} K^{-1}$)
C_{v}	volumetric heat capacity (J $m^{-3} K^{-1}$)
CRI	color rendering index
D_{nh}	dark signal for normal-hemispherical transmittance measurement (%)
е	thermal effusivity (J m ^{-2} K ^{-1} s ^{$-1/2$})
Ε	Young's modulus (GPa)
ΔE	geometrical distance in the CIE 1964 uniform chromaticity space
h	haze (%)
Н	hardness (GPa)
Ι	intensity (a.u.)
k	thermal conductivity (W m ^{-1} K ^{-1})
Μ	molar mass (g mol ⁻¹)
M_d	reference signal for diffuse transmittance measurement (%)
Ν	number of points
p	pressure (Pa)
p_0	saturation pressure (Pa)
R_a	average surface roughness (nm)
R_{RMS}	root mean squared surface roughness (nm)
S_{BET}	specific surface area (m ² g ⁻¹)
S_d	diffuse transmitted signal (%)
S_{nh}	normal-hemispherical transmitted signal (%)
ī	average thickness (mm)

 T_{nh} normal-hemispherical transmittance (%)

 V_{mi} micropore volume (cm³ g⁻¹)

 V_t total pore volume (cm³ g⁻¹)

w weight fraction

 w_p pore width (nm)

 W^* , u^* , v^* trichromatic coordinates in the CIE 1964 uniform color space domain

z vertical stylus displacement (nm)

Greek symbols

α_s	reduced adsorption
λ	wavelength (nm)
ρ	density (g cm ^{-3})
σ	surface density of surface groups (mol m ⁻²)
ϕ	porosity or volume fraction

Subscripts and superscripts

ave	refers to average
ave	
eff	refers to effective
exp	refers to experimental
fit	refers to values retrieved from fit
H ₂ O	refers to water
ref	refers to reference material
OH	refers to surface hydroxyl groups
SiO ₂	refers to silica

S1. Materials and methods

Chemicals. Colloidal solution of silica nanoparticles Nalco 2326 (15 wt% in water, NH₃ stabilized, lot number BP7J1239A1) was purchased from Nalco Chemical Company (Naperville, IL, USA). Perfluoropolyether oils Krytox GPL 100, GPL 104, and GPL 106 with general formula $[CF(CF_3)CF_2O]_n$ where n = 10-60, used as the liquid substrates, were purchased from Miller-Stephenson Chemical Company Inc. (Danbury, CT, USA). All chemicals were used as received without further purification.

Material synthesis. The exact synthesis of nanoparticle-based mesoporous silica slabs on PFC was as follows: between 5 and 30 mL of the colloidal solution of silica nanoparticles was placed in a PTFE mold with PFC liquid (either Krytox GLP 100, GPL 104 or GLP 106) covering the bottom

surface in a 1–3 mm thick layer. Water was slowly removed by evaporation resulting in gelation and aging of nanoparticle-based slabs in the shape of the mold. The slabs were then slowly dried to remove all water without cracking leading to mesoporous silica slabs. The drying rate was controlled via (i) the mold's opening size, (ii) temperature, and (iii) surrounding relative humidity. Here, the samples were dried either (i) in ambient air at room temperature or in a convection oven at 25 °C, with a mold's opening corresponding to 0–10% of the mold surface area, (ii) in a convection oven at 40 °C with a mold's opening completely covered, or (iii) in a humidity chamber at room temperature and relative humidity of 50–80%. The resulting mesoporous silica slabs were calcined in oxygen at 400 °C for 2 h using a 5 °C min⁻¹ temperature ramp to remove any NH₃ and PFC residues.

The synthesis of mesoporous silica slabs on a solid PTFE substrate was the same as that described above except that the colloidal solution of silica nanoparticles was placed in an empty PTFE mold without PFC liquid. In addition, the mesoporous slabs prepared on PTFE were analyzed as synthesized, i.e., without calcination.

S2. Characterization methods

Structural Characterization. The average thickness \bar{t} (in mm) of nanoparticle-based mesoporous silica slabs was estimated by averaging the maximum and minimum thickness of a slab measured using calipers. This was done to accurately represent thickness of the slabs for the purpose of comparing their optical characteristics since some slabs gradually thickened or thinned toward the edges while some slabs remained completely flat toward the edges..

Transmission electron microscope (TEM) images of mesoporous silica slabs were taken using a 200 kV FEI Tecnai TF20 TEM equipped with a field emission gun. The samples were first grounded, then suspended in ethanol and sonicated for 3 min, and applied onto a 400-mesh carboncoated copper TEM grid. The resulting TEM specimens were air dried before TEM analysis. Diameters of at least 50 distinct silica nanoparticles were manually measured using ImageJ¹ software to calculate average diameter and standard deviation of the silica nanoparticles.

Low-temperature nitrogen adsorption–desorption isotherms were measured at –196 °C using a surface area and porosity analyzer TriStar II 3020 (Micromeritics Instrument Corp., Norcross, GA, USA). Each sample was degassed in vacuum at 150–200 °C for 20–24 h prior to measurements. The specific surface area S_{BET} (in m² g⁻¹) was calculated using the Brunauer– Emmett–Teller method² (BET) based on low-temperature nitrogen adsorption data in the relative pressure range $p/p_0 = 0.05-0.2$ and assuming the cross-sectional area of a nitrogen molecule to be 0.162 nm².³ The total specific pore volume V_t (in cm³ g⁻¹) was calculated by converting the number of moles of nitrogen adsorbed at a relative pressure $p/p_0 = 0.98$ to the volume of liquid nitrogen based on the liquid nitrogen molar density of 34.38 cm³ mol⁻¹.⁴ Then, the porosity ϕ was calculated from the total specific pore volume V_t according to⁵

$$\phi = \frac{V_t \,\rho_{SiO_2}}{1 + V_t \,\rho_{SiO_2}}.\tag{S1}$$

The micropore volume V_{mi} (in cm³ g⁻¹) was calculated using the α_s -comparative method⁵ in the reduced adsorption range $\alpha_s = 0.4$ –0.8 and using macroporous silica LiChrospher Si-1000 as a reference.⁶ The reduced adsorption α_s was defined as⁵

$$\alpha_s = a_{ref} / a_{ref,0.4},\tag{S2}$$

where a_{ref} and $a_{ref,0.4}$ are the volume of nitrogen adsorbed on the reference material surface at a given relative pressure p/p_0 and at $p/p_0 = 0.4$, respectively. In addition, the peak pore width w_p (in nm) was estimated from the position of peak maximum in the pore size distribution determined using the Kruk–Jaroniec–Sayari (KJS) method⁷ based on the Barrett–Joyner–Halenda (BJH) method⁸ using (i) the adsorption branch of nitrogen isotherm, (ii) the modified Kelvin equation⁷

calibrated for cylindrical pores up to 19 nm in diameter, and (iii) the statistical film thickness curve derived from the nitrogen adsorption isotherm measured for macroporous silica LiChrospher Si-1000.⁶

Two-dimensional small-angle X-ray scattering data were collected at the Stanford Synchrotron Lightsource (SSRL) using beamlines 1–5 with a wavelength of 0.1033 nm operated at an X-Ray energy of 12.002 keV and detector distance of 2.870 m using a Rayonix-165 CCD detector. The data was then calibrated and reduced using the Nika package from Igor Pro.⁹ The reduced small-angle X-ray scattering (SAXS) patterns were fit using (a) unified model^{10,11} implementing correction to the Porod scaling factor¹² and (b) pseudo-Voigt peaks by minimizing average relative error

$$\left(\frac{\Delta I}{I}\right)_{ave} = \frac{1}{N} \sum_{i=1}^{N} \frac{\left|I_{exp,i} - I_{fit,i}\right|}{I_{exp,i}} \tag{S3}$$

Pauly et al.¹³ showed that worm-like mesopores can yield up to two weak correlation peaks while remaining disordered. The contribution of these weak correlation peaks was subtracted from SAXS data to accurately calculate slopes of Porod plots.

Optical Characterization. The spectral normal-hemispherical transmittance $T_{nh,\lambda}$ and spectral diffuse transmittance $T_{d,\lambda}$ of the mesoporous silica slabs, at wavelength between 400 and 800 nm, were measured with a double-beam UV-Vis spectrometer 3101-PC (Shimadzu, Kyoto, Japan) equipped with an integrating sphere ISR 3100 (Shimadzu, Kyoto, Japan) with internal diameter of 6 cm. The measured transmitted signal $S_{nh,\lambda}$ was corrected for the reference signal $B_{nh,\lambda}$ and the dark signal $D_{nh,\lambda}$ according to

$$T_{nh,\lambda} = \frac{S_{nh,\lambda} - D_{nh,\lambda}}{B_{nh,\lambda} - D_{nh,\lambda}}.$$
(S4)

Supporting Information

The dark signal $D_{nh,\lambda}$ was collected in the absence of light while the reference signal $B_{nh,\lambda}$ was measured using the same procedure as that used to measure $S_{nh,\lambda}$ but in absence of sample. In addition, the diffuse transmitted signal $S_{d,\lambda}$ was measured by removing the cover of the back port of the integrating sphere opposite to the sample port so that the normal-normal beam of light left the integrating sphere while the scattered light transmitted through the sample was collected and measured. Moreover, haze h_{λ} was estimated according to ASTM D1003-13 standard as¹⁴

$$h_{\lambda} = \frac{S_{d,\lambda}}{S_{nh,\lambda}} - \frac{M_{d,\lambda}}{B_{nh,\lambda}}.$$
(S5)

where $M_{d,\lambda}$ was the signal measured using the same procedure as that used to measure $S_{d,\lambda}$ but without any sample present.

The color rendering index *CRI* of the slabs quantifies the quality of color transmission through semitransparent materials. It ranges from 0 to 100, with *CRI* \approx 100 indicating excellent color reproduction and *CRI* \geq 95 being acceptable.¹⁵ Most glass and windows have color rendering index *CRI* = 80–96.¹⁶ The CRI of mesoporous silica slabs was calculated according to BS EN 410:2011 standard¹⁷ taking the CIE (International Commission on Illumination) illuminant D_{65} as the light source. First, the trichromatic coordinates W_i^* , u_i^* , and v_i^* , in the CIE 1964 uniform color space domain, of eight test colors were calculated from the spectral normal-hemispherical transmittance $T_{nh,\lambda}$. Then, the geometrical distance ΔE_i , in the CIE 1964 uniform chromaticity space, between the color perceived with and without a sample was calculated for each of the eight test colors as¹⁷

$$\Delta E_i = \sqrt{\Delta W_i^{*2} + \Delta u_i^{*2} + \Delta v_i^{*2}} \quad \text{with } i = 1, 2, ..., 8.$$
 (S6)

Here, the trichromatic coordinates W_i^* , u_i^* , and v_i^* of the reference illuminant D_{65} were taken from BS EN 410:2011 standard.¹⁷ Finally, the color rendering index *CRI* was calculated as¹⁷

Supporting Information

$$CRI = \frac{1}{8} \sum_{i=1}^{8} (100 - 4.6\Delta E_i).$$
(S7)

The average R_a and root mean squared R_{RMS} surface roughness values of the mesoporous silica slabs were measured using a surface profilometer Dektak 6 (Veeco, Plainview, NY, USA) according to¹⁸

$$R_a = \frac{1}{N} \sum_{i=1}^{N} |\bar{z} - z_i| \text{ and } R_{RMS} = \sqrt{\frac{1}{N} \sum_{i=1}^{N} (\bar{z} - z_i)^2},$$
(S8)

where *N* is the total number of vertical stylus displacement measurements, z_i is the vertical stylus displacement at the *i*th equally spaced location along a line trace on the sample's surface, and \bar{z} is the average vertical stylus displacement.

Thermal Characterization. Effective thermal conductivity k_{eff} of the mesoporous slabs was measured using the time-domain thermoreflectance (TDTR) method.^{19–21} It was calculated from the measured effective thermal effusivity e_{eff} of mesoporous silica slabs as¹⁹

$$k_{eff} = \frac{e_{eff}^2}{c_{v,eff}} = \frac{e_{eff}^2}{\rho_{eff}c_{p,eff}} = \frac{e_{eff}^2}{\phi_{sio_2}\rho_{sio_2}c_{p,eff}},$$
(S9)

where $c_{v,eff}$ is the effective volumetric heat capacity (in J m³ K⁻¹) where ρ_{eff} , $c_{p,eff}$, and $\phi_{SiO_2} = 1 - \phi$ are the effective density (in kg m⁻³), effective specific heat capacity (in J kg⁻¹ K⁻¹), and silica volume fraction of the mesoporous silica slabs, respectively. First, the sample was coated with a 80-nm thick Al film. Second, the Al-coated sample was heated to 160 °C for 1 hour at a pressure < 1 Pa in a cryostat to remove all physisorbed water, and then allowed to cool to room temperature (note that only SiO₂-PTFE-rt sample was degassed while SiO₂-PFC-hc sample was measured as prepared). Third, 2-fs laser pulses with wavelengths of 400 and 800 nm were used to generate heat (pump beam) and to detect temperature (probe beam), respectively. The pump and probe laser spot were 20 µm in diameter to average the reflected signal over a surface area much larger than the average cross-sectional area of a pore and to measure the effective thermal

effusivity of the porous sample. Here, the metallic film was used (i) as a transducer converting the photon energy of the pump laser pulse into thermal energy and (ii) as a sensor whose reflectance of the probe beam depends on temperature. After heating with the pump beam pulse, the temporal temperature decay of the Al film was recorded with the probe beam based on a linear relationship between reflectivity and temperature of the metallic film. Finally, the transient temperature decay curve was fitted with a multilayer heat conduction model to obtain the effective thermal effusivity.²⁰ Measurements were performed at least at five different locations across the sample and averaged.

The effective specific heat capacity $c_{p,eff}$ of degassed SiO₂-PTFE-rt sample at room temperature was estimated by accounting for the presence of hydroxyl groups at the pore walls according to²²

$$c_{p,eff} = c_{p,SiO_2} + M_{OH}\sigma_{OH}S_{BET}(c_{p,OH} - c_{p,SiO_2}),$$
(S10)

where $c_{p,SiO_2} = 0.733 \text{ J g}^{-1} \text{ K}^{-1}$ is the specific heat capacity of bulk amorphous silica,²³ $M_{OH} = 17$ g mol⁻¹ is the molar mass of OH groups, $\sigma_{OH} = 4.9 \text{ OH nm}^{-2}$ is the surface density of OH groups,²⁴ and $c_{p,OH} = 1.58 \text{ J g}^{-1} \text{ K}^{-1}$ is the specific heat capacity of OH groups.²⁵ Note that silica has a strong affinity for water and mesoporous silica offers large pore volume and surface area for water to condense in the mesopores. Thus, the effective specific heat capacity $c_{p,eff}^*$ of the SiO₂-PFC-hc sample under ambient conditions with non-zero relative humidity was estimated as

$$c_{p,eff}^* = (1 - w_{H_20})c_{p,eff} + w_{H_20}c_{p,H_20},$$
(S11)

where $w_{H_20} = 0.05$ is the estimated weight fraction of water adsorbed on the slab surface, $c_{p,eff}$ is the effective specific heat capacity of degassed mesoporous silica given by Equation (S10), and $c_{p,H_20} = 4.181 \text{ J g}^{-1} \text{ K}^{-1}$ is the specific heat capacity of water at 25 °C.²⁶

Supporting Information

The combined relative uncertainty at 95% confidence level of the effective thermal conductivity was calculated as

$$\frac{\Delta k_{eff}}{k_{eff}} = \sqrt{\left(\frac{\Delta e_{eff}^2}{e_{eff}^2}\right)^2 + \left(\frac{\Delta \phi_{SiO_2}}{\phi_{SiO_2}}\right)^2 + \left(\frac{\Delta c_{p,eff}}{c_{p,eff}}\right)^2},\tag{S12}$$

where Δe_{eff}^2 was estimated as a standard deviation of a mean value at 95% confidence level and $\Delta \phi_{SiO_2} = \Delta \phi = \pm 0.02$ was estimated based on prior experience. Uncertainty of the effective specific heat capacity was estimated as

$$\Delta c_{p,eff} = M_{OH} \sigma_{OH} (c_{p,OH} - c_{p,SiO_2}) \Delta S_{BET} \text{ and}$$

$$\Delta c_{p,eff}^* = \sqrt{\left[\left(1 - w_{H_2O} \right) \Delta c_{p,eff} \right]^2 + \left[\left(c_{p,H_2O} - c_{p,eff} \right) \Delta w_{H_2O} \right]^2} \tag{S13}$$

where $\Delta S_{BET} = 0.1 S_{BET}$.²²

Mechanical Characterization. Effective Young's modulus E_{eff} and hardness H_{eff} of the mesoporous silica slabs were measured using a MTS Nanoindenter XP instrument (MTS Nano Instruments Inc., Oak Ridge, TN, USA) equipped with a diamond Berkovich pyramidal tip. The continuous stiffness measurement (CSM) method was used to measure Young's modulus and hardness of the mesoporous slabs as a function of displacement that were then averaged over the displacement range 500–2000 nm. Each sample was indented at 16 different locations arranged in a 4 x 4 grid with 25 µm wide pitch. The sample-average Young's modulus and hardness were calculated by averaging all valid indentation-average values. All samples were indented using the harmonic displacement of 2 nm, frequency of 45 Hz, strain rate of 0.05 sec⁻¹, and depth limit of 2000 nm. Poisson's ratio was taken as 0.20.^{27–29}



Figure S1. Optical images of bubbles-free transparent nanoparticle-based mesoporous silica monolithic slabs prepared on defect-free PTFE at room temperature. Both slabs had diameter of ≈2.5 cm and average thickness ≈2 mm.



Figure S2. Optical images of (a) bottom surface of a SiO₂-PFC slab with smooth surface from the PFC liquid (few visible scratches and defects are due to handling of the slab) and (b) bottom surface of a SiO₂-PTFE slab with noticeable surface roughness from the PTFE mold.



Figure S3. Optical images of (a) nanoparticle-based mesoporous silica monolithic slab with bubbles on the bottom surface prepared on PTFE at room temperature (diameter 4.5 cm and average thickness 3.5 mm) and (b) bubbles at the bottom of a SiO₂-PTFE-rt slab formed along a scratch in PTFE (also imprinted in the slab).

References

- (1) Schneider, C. A.; Rasband, W. S.; Eliceiri, K. W. NIH Image to ImageJ: 25 Years of Image Analysis. *Nat. Methods* **2012**, *9*, 671–675. https://doi.org/10.1038/nmeth.2089.
- (2) Brunauer, S.; Emmett, P. H.; Teller, E. Adsorption of Gases in Multimolecular Layers. J. Am. Chem. Soc. **1938**, 60 (2), 309–319. https://doi.org/10.1021/ja01269a023.
- (3) Kruk, M.; Jaroniec, M. Gas Adsorption Characterization of Ordered Organic-Inorganic Nanocomposite Materials. *Chem. Mater.* **2001**, *13* (10), 3169–3183. https://doi.org/10.1021/cm0101069.
- (4) Naono, H.; Hakuman, M.; Shiono, T. Analysis of Nitrogen Adsorption Isotherms for a Series of Porous Silicas with Uniform and Cylindrical Pores: A New Method of Calculating Pore Size Distribution of Pore Radius 1–2 Nm. J. Colloid Interface Sci. 1997, 186 (2), 360– 368. https://doi.org/10.1006/jcis.1996.4677.
- (5) Rouquerol, J.; Rouquerol, F.; Llewellyn, P.; Maurin, G.; Sing, K. S. W. Adsorption by *Powders and Porous Solids: Principles, Methodology and Applications*, 2 edition.; Academic Press: Amsterdam, 2013.
- (6) Jaroniec, M.; Kruk, M.; Olivier, J. P. Standard Nitrogen Adsorption Data for Characterization of Nanoporous Silicas. *Langmuir* 1999, 15 (16), 5410–5413. https://doi.org/10.1021/la990136e.
- Kruk, M.; Jaroniec, M.; Sayari, A. Application of Large Pore MCM-41 Molecular Sieves To Improve Pore Size Analysis Using Nitrogen Adsorption Measurements. *Langmuir* 1997, *13* (23), 6267–6273. https://doi.org/10.1021/la970776m.
- (8) Barrett, E. P.; Joyner, L. G.; Halenda, P. P. The Determination of Pore Volume and Area Distributions in Porous Substances. I. Computations from Nitrogen Isotherms. J. Am. Chem. Soc. 1951, 73 (1), 373–380. https://doi.org/10.1021/ja01145a126.
- (9) Ilavsky, J. Nika: Software for Two-Dimensional Data Reduction. J. Appl. Crystallogr. 2012, 45 (2), 324–328. https://doi.org/10.1107/S0021889812004037.
- Beaucage, G. Approximations Leading to a Unified Exponential/Power-Law Approach to Small-Angle Scattering. J. Appl. Crystallogr. 1995, 28 (6), 717–728. https://doi.org/10.1107/S0021889895005292.
- (11) Beaucage, G. Small-Angle Scattering from Polymeric Mass Fractals of Arbitrary Mass-Fractal Dimension. J. Appl. Crystallogr. **1996**, 29 (2), 134–146. https://doi.org/10.1107/S0021889895011605.
- (12) Hammouda, B. Analysis of the Beaucage Model. J. Appl. Crystallogr. **2010**, 43 (6), 1474–1478. https://doi.org/10.1107/S0021889810033856.
- (13) Pauly, T. R.; Liu, Y.; Pinnavaia, T. J.; Billinge, S. J. L.; Rieker, T. P. Textural Mesoporosity and the Catalytic Activity of Mesoporous Molecular Sieves with Wormhole Framework Structures. J. Am. Chem. Soc. 1999, 121 (38), 8835–8842. https://doi.org/10.1021/ja991400t.
- (14) Standard Test Method for Haze and Luminous Transmittance of Transparent Plastics (ASTM D1003—13). ASTM International.
- (15) Gunde, M. K.; Krašovec, U. O.; Platzer, W. J. Color Rendering Properties of Interior Lighting Influenced by a Switchable Window. *JOSA A* 2005, *22* (3), 416–423. https://doi.org/10.1364/JOSAA.22.000416.
- (16) Dangol, R.; Kruisselbrink, T.; Rosemann, A. Effect of Window Glazing on Colour Quality of Transmitted Daylight. J. Daylighting 2017, 4 (2), 37–47. https://doi.org/10.15627/jd.2017.6.

- (17) Glass in Building. Determination of Luminous and Solar Characteristics of Glazing (BS EN 410:2011). British Standards Institution.
- (18) Den Outer, A.; Kaashoek, J. F.; Hack, H. R. G. K. Difficulties with Using Continuous Fractal Theory for Discontinuity Surfaces. *Int. J. Rock Mech. Min. Sci. Geomech. Abstr.* 1995, 32 (1), 3–9. https://doi.org/10.1016/0148-9062(94)00025-X.
- (19) Hu, Y.; Zeng, L.; Minnich, A. J.; Dresselhaus, M. S.; Chen, G. Spectral Mapping of Thermal Conductivity through Nanoscale Ballistic Transport. *Nat. Nanotechnol.* 2015, *10* (8), 701– 706. https://doi.org/10.1038/nnano.2015.109.
- (20) Cahill, D. G. Analysis of Heat Flow in Layered Structures for Time-Domain Thermoreflectance. *Rev. Sci. Instrum.* **2004**, *75* (12), 5119–5122. https://doi.org/10.1063/1.1819431.
- (21) Schmidt, A. J.; Chen, X.; Chen, G. Pulse Accumulation, Radial Heat Conduction, and Anisotropic Thermal Conductivity in Pump-Probe Transient Thermoreflectance. *Rev. Sci. Instrum.* 2008, 79 (11), 114902. https://doi.org/10.1063/1.3006335.
- Marszewski, M.; Butts, D.; Lan, E.; Yan, Y.; King, S. C.; McNeil, P. E.; Galy, T.; Dunn, B.; Tolbert, S. H.; Hu, Y.; et al. Effect of Surface Hydroxyl Groups on Heat Capacity of Mesoporous Silica. *Appl. Phys. Lett.* 2018, *112* (20), 201903. https://doi.org/10.1063/1.5027080.
- (23) Richet, P.; Bottinga, Y.; Denielou, L.; Petitet, J. P.; Tequi, C. Thermodynamic Properties of Quartz, Cristobalite and Amorphous SiO2: Drop Calorimetry Measurements between 1000 and 1800 K and a Review from 0 to 2000 K. *Geochim. Cosmochim. Acta* 1982, 46 (12), 2639–2658. https://doi.org/10.1016/0016-7037(82)90383-0.
- (24) Zhuravlev, L. T. Concentration of Hydroxyl Groups on the Surface of Amorphous Silicas. *Langmuir* **1987**, *3* (3), 316–318. https://doi.org/10.1021/la00075a004.
- (25) Rihani, D. N.; Doraiswamy, L. K. Estimation of Heat Capacity of Organic Compounds from Group Contributions. *Ind. Eng. Chem. Fundam.* **1965**, *4* (1), 17–21. https://doi.org/10.1021/i160013a003.
- (26) *CRC Handbook of Chemistry and Physics*, 98th edition 2017-2018.; Rumble, J. R., Ed.; CRC Press, Taylor & Francis Group: Boca Raton London New York, 2017.
- (27) Gross, J.; Reichenauer, G.; Fricke, J. Mechanical Properties of SiO 2 Aerogels. J. Phys. Appl. Phys. **1988**, 21 (9), 1447. https://doi.org/10.1088/0022-3727/21/9/020.
- (28) Moner-Girona, M.; Roig, A.; Molins, E.; Martínez, E.; Esteve, J. Micromechanical Properties of Silica Aerogels. *Appl. Phys. Lett.* 1999, 75 (5), 653–655. https://doi.org/10.1063/1.124471.
- (29) Patil, S. P.; Rege, A.; Sagardas; Itskov, M.; Markert, B. Mechanics of Nanostructured Porous Silica Aerogel Resulting from Molecular Dynamics Simulations. J. Phys. Chem. B 2017, 121 (22), 5660–5668. https://doi.org/10.1021/acs.jpcb.7b03184.